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Structurally ferroelectric SrMgF₄

The crystal structure of 0.06% Ce-doped SrMgF₄, strontium magnesium tetrafluoride, reported by Ishizawa *et al.* [(2001), *Acta Cryst.* C57, 784–786] is shown to satisfy the structural criteria for ferroelectricity and to have a predicted Curie temperature $T_c \simeq 1450$ K. The estimated spontaneous polarization $P_s \simeq 11 \times 10^{-2}$ C m⁻² is consistent with classification as a two-dimensional ferroelectric in which minor Δx and major Δy , Δz atomic coordinate component displacements are required for ferroelectric switching.

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1. Introduction

All members of the $BaMF_4$ family (M = Mg, Mn, Fe, Co, Ni and Zn) were reported by Eibschütz et al. (1969) to exhibit ferroelectricity, each with a spontaneous polarization (P_s) between ~ 7 and $10 \times 10^{-2} \text{ Cm}^{-2}$;¹ Keve *et al.* (1969, 1970) determined the structure of the family members M = Mn and Co. The structures of BaMgF₄, BaFeF₄, BaNiF₄ and BaZnF₄, isostructural with BaMnF₄ and BaCoF₄, were subsequently determined by Gingl (1997), Averdunk & Hoppe (1988), Cox et al. (1970), von Schnering & Bleckmann (1968) and Lapasset et al. (1996), respectively. The recent redetermination of the structure of $SrMgF_4$ by Ishizawa et al. (2001) raised the possibility, by analogy with the BaMF₄ family, that it also might be ferroelectric, although it crystallizes in space group $P112_1$, whereas BaMgF₄ forms in space group $A2_1am(Cmc2_1)$. SrMgF₄ was first prepared by Banks *et al.* (1980) who identified it as orthorhombic, with lattice constants comparable to those of BaMnF₄. A negative test for second harmonic generation led them to suggest the space group Amam; they also showed that solid solutions with $\sim 1\%$ EuMgF₄ were photoluminescent. Apart from Banks et al.'s (1980) observation that $SrMgF_4$ melts incongruently at ~1175 K and Bingyi & Banks (1982) report of a solid-state transformation in the SrF₂-MgF₂ binary system at 1080 (10) K, no additional physical measurements were found in the literature other than spectroscopic, on rare-earth doped SrMgF₄, e.g. Wu & Shi (1995, 1996) and Sun et al. (1995). Ishizawa et al. (2001) have now shown that SrMgF₄ forms a monoclinic superstructure of the earlier orthorhombic arrangement. Analysis of the monoclinic coordinates obtained thereby shows that they fully satisfy the structural criteria for ferroelectricity.

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Table 1

Atomic coordinates of SrMgF₄ at 298 K in space group $P112_1$ (Ishizawa *et al.*, 2001), hypothetical x'y'z' coordinates in supergroup $P112_1/m$, atomic displacements (Δx , Δy , Δz) in Å at the phase transition and thermal/static (u_{iso}) displacement in Å at 298 K.

$a = 7.8249$ (8), $b = 7.4930$ (7), $c = 16.9248$ (17) Å, $\gamma = 105.041$ (11)°.	$z^* = z - 0.270842; \Delta x = (x - x')a, \Delta y = (y - y')b, \Delta z = (z^* - z')$	IC.
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	Wyckoff position P112 ₁ ,										
	$P112_{1}/m$	x	у	<i>z</i> *	<i>x</i> ′	y'	<i>z</i> ′	Δx	Δy	Δz	$u_{\rm iso}$
Sr1	$2(a), 2(e)^{\dagger}$	0.18015 (12)	0.2084 (2)	0.24503 (7)	0.18015	0.2084	1/4	0	0	-0.084	0.12 (2)
Sr2	2(a) 4(f)	0.33110 (9)	0.8164 (2)	0.40647 (9)	0.32531	0.79732	0.41321	0.045	0.143	-0.114	0.11 (2)
Sr6	2(a)	0.31952 (8)	0.7783 (1)	0.08005 (7)	0.32531	0.7973	0.08679	-0.045	-0.143	-0.114	0.11(2)
Sr3	$2(a) \\ 4(f)$	0.83449 (8)	0.8308 (1)	0.06336 (8)	0.83480	0.8336	0.08204	-0.002	-0.022	-0.316	0.09 (2)
Sr5	2(a)	0.83511 (9)	0.8365 (2)	0.39929 (8)	0.83480	0.8336	0.41797	0.002	0.022	-0.316	0.11 (2)
Sr4	2(a), 2(e)	0.66509 (10)	0.1619 (1)	0.22916	0.66509	0.1619	1/4	0	0	-0.353	0.09 (2)
Mg1	2(a), 2(e)	0.5423 (3)	0.6592 (4)	0.2521 (2)	0.5423	0.6592	1/4	0	0	0.036	0.09 (3)
Mg2	$2(a) \\ 4(f)$	0.5403 (3)	0.6683 (3)	-0.0801 (2)	0.5418	0.6693	-0.0823	-0.012	-0.007	0.037	0.09 (3)
Mg3	2(a)	0.4568 (3)	0.3296 (3)	0.0845(2)	0.4582	0.3307	0.0823	-0.012	-0.008	0.037	0.08(3)
Mg4	2(a), 2(e)	0.0379 (3)	0.6634 (4)	0.2520 (2)	0.0379	0.6634	1/4	0	0	0.034	0.09 (3)
Mg5	2(a)	0.0432 (3)	0.6633 (3)	-0.0804 (2)	0.0427	0.6672	-0.0834	-0.005	-0.029	0.050	0.08 (3)
Mg6	2(a)	0.9579 (3)	0.3290 (3)	0.0863(2)	0.9573	0.3328	0.0834	-0.005	-0.028	0.050	0.08(3)
F1	2(a)	0.5964 (5)	0.8285 (6)	0.1606 (3)	0.5935	0.8591	0.1720	0.023	-0.229	-0.193	0.11 (4)
F3	2(a)	0 5905 (6)	0.8897(5)	0.3166 (3)	0 5935	0.8591	0 3280	-0.023	0.229	-0.193	0.12(4)
F2	2(a)	0.5008 (6)	0.4286 (5)	0.1952 (3)	0.4879	0.4903	0.1684	0.101	-0.462	0.454	0.11 (4)
F4	2(a)	0.4749 (6)	0.5519 (5)	0.3585 (3)	0.4879	0.4903	0.3316	-0.101	0.462	0.455	0.14 (4)
F5	2(a)	0.5656 (6)	0.8222 (6)	-0.1750 (3)	0.5954	0.8644	-0.1614	-0.233	-0.317	-0.230	0.12 (4)
F9	2(a)	0.3749 (5)	0.0934(5)	0.1478 (3)	0.4046	0.1356	0.1614	-0.232	-0.317	-0.230	0.11 (4)
F6	2(a)	0.5779 (6)	0.8994 (5)	-0.0147 (3)	0.5833	0.8479	-0.0002	-0.042	0.386	-0.246	0.11 (4)
F8	2(a)	0.4114 (6)	0.2036 (6)	-0.0144(3)	0.4167	0.1521	0.0002	-0.041	0.386	-0.246	0.13 (4)
F7	2(a), 2(d)	0.5463 (5)	0.5698(5)	0.0320(2)	1/2	1/2	0	0.362	0.523	0.542	0.11 (4)
F10	2(a), 2(e)	0.7898 (5)	0.6663(7)	0.2710(3)	0.7898	0.6663	1/4	0	0	0.355	0.11 (4)
F11	2(a)	0.2826 (5)	0.6224 (6)	-0.0666(3)	0.2932	0.6676	-0.0772	-0.084	-0.339	0.179	0.13 (4)
F24	2(a)	0.6963 (5)	0.2872(6)	0.0878(3)	0 7068	0 3324	0.0772	-0.083	-0.339	0 179	0.13(4)
F12	2(a)	0.2190 (5)	0.3753 (6)	0.0962 (3)	0.2076	0.3296	0.0889	0.090	0.343	0.124	0.13 (4)
F23	2(a)	0.8038 (5)	0.7162 (6)	-0.0815(3)	0 7924	0.6704	-0.0889	0.090	0 343	0.125	0.12(4)
F13	2(a) 2(a)	0.0584 (5)	0.7965 (6)	0.1540 (3)	0.0839	0.8507	0.1707	-0.201	-0.406	-0.282	0.12 (4)
F15	2(a)	0.1094(5)	0.9048(5)	0.3127 (3)	0.0839	0.8507	0.3293	0.201	0.406	-0.282	0.11 (4)
F14	2(a)	0.9613 (6)	-0.5748(5)	0.1992 (3)	1.0040	-0.5050	0.1686	0.334	-0.523	0.518	0.12 (4)
F16	2(a)	0.9534(5)	0.4351 (5)	-0.1379(3)	0.9960	0.5050	-0.1686	-0.333	-0.524	0.519	0.12 (4)
F17	2(a)	0.0965 (5)	0.8283 (5)	-0.1736(3)	0.0860	0.8645	-0.1624	0.083	-0.271	-0.190	0.12 (1)
F21	2(a)	0.9246 (5)	0.0993(5)	0.1512 (3)	0.9140	0.1355	0.1624	0.084	-0.271	-0.190	0.09 (4)
F18	2(a)	0.1196 (6)	0.8921 (5)	-0.0143(3)	0.0961	0.8654	-0.0036	0.185	0.200	-0.181	0.11 (4)
F20	2(a)	0.9274 (6)	0.1613 (6)	-0.0071(3)	0.9039	0.1346	0.0036	0.185	0.200	-0.181	0.11 (4)
F19	2(a), 2(b)	0.9808(5)	0.5533(5)	0.0262(3)	0	1/2	0	-0.151	0.399	0.443	0.12 (4)
F22	2(a), 2(e)	0.2901 (6)	0.6563 (7)	0.2393 (3)	0.2901	0.6563	1/4	0	0	-0.181	0.13 (4)

† Equivalent 2(a) positions in space group P112₁, x, y, z; \bar{x} , \bar{y} , $\frac{1}{2}$ + z. In P112₁/m for 4(f), x, y, z; \bar{x} , \bar{y} , $\frac{1}{2}$ + z; \bar{x} , \bar{y} , \bar{z} ; x, y, $\frac{1}{2}$ - z. For 2(e), x, y, $\frac{1}{4}$; \bar{x} , \bar{y} , $\frac{3}{4}$; for 2(d), $\frac{1}{2}$, $\frac{1}{2}$, $\frac{1}{2}$, $\frac{1}{2}$; and for 2(b), 0, $\frac{1}{2}$, $\frac{1}{2}$, $\frac{1}{2}$, $\frac{1}{2}$, $\frac{1}{2}$; $\frac{1$

2. Atomic coordinates, structural criteria for ferroelectricity and predicted Curie temperature of SrMgF₄

The unusual displacement of all Mg atoms from the centers of their F-atom octahedra in $SrMgF_4$ resulting from the proposed space group *Amam* led Ishizawa *et al.* (2001) to redetermine the atomic arrangement. Refinement of 326 parameters in

space group $P112_1$ was undertaken using 4130 independent absorption- and extinction-corrected Friedel pairs measured at 298 K with $I > 3\sigma(I)$; 239 reflections were excluded. The composition of the Ce-doped Bridgman-grown crystal, Sr:Ce = 0.9994:0.0006, was assumed equivalent to stoichiometric in the structure analysis. The final values of R = 0.044, wR = 0.037 are given for the xyz atomic coordinates presented in Table 1, in which an origin shift of -0.27084 has been applied along the polar axis. This shift minimizes the total $\Sigma \Delta z$, see below, for a transition to a prototype structure with the supergroup assumed to be $P112_1/m$. Alternative supergroups may be possible, but are not expected to change the present conclusions significantly; experimental investigation at higher temperatures would be appropriate. Table 1 additionally gives the set of x'y'z' coordinates in the prototype structure derived from the xyz values under the requirement of conformity with the symmetry of Wyckoff positions 4(f), 2(e), 2(d) or 2(b) in space group $P112_1/m$, also the corresponding $x - x' = \Delta x$, $y - y' = \Delta y$, $z - z' = \Delta z$ axial displacements. Table 1 shows that no atom in space group $P112_1/m$ in the predicted phase I.

The Δz magnitudes in Table 1 fully satisfy the structural criteria for ferroelectricity (Abrahams, 2000), namely:

(a) that the *i*th atom forming the shortest and least-ionic bonds in the unit cell (Mg in the case of SrMgF₄) undergoes no polar displacement $\Delta z_l \gtrsim 1$ Å between the location reported at $x_i y_i z_i$ and that at which the resulting spontaneous polarization P_s vanishes, and

(b) that the largest Δz_i exceeds ~0.1 Å or the r.m.s. thermal or static displacement u of that *i*th atom.

 $SrMgF_4$ is hence confidently predicted to be a new ferroelectric crystal.

The Curie temperature T_c of a ferroelectric with known crystal structure may also be predicted if the *i*th atom therein occupies an octahedron of O or F atoms, as in the present case. Abrahams *et al.* (1968) have shown, for such structures, that

$$T_c = (\kappa/2k)(\Delta z_i)^2 \,\mathrm{K},\tag{1}$$

where the force constant $\kappa \simeq 5.52$ kPa, k is Boltzmann's constant, Δz_i is the largest displacement along the polar c axis by the *i*th atom defined above and $\kappa/2k = 2.00$ (9) $\times 10^4$ K Å⁻².

The six independent MgF₆ octahedra in SrMgF₄ phase II may be seen in Table 1 to undergo displacement at the transition to phase I by forming three symmetry-related pairs of octahedra. The Mg1 and Mg4 pair occupies the mirror plane at $z = \frac{1}{4}$, the Mg2, Mg3 and Mg5, Mg6 pairs in Table 1 becoming related by an inversion center. However, symmetry equivalents of Mg3 and Mg6 in Table 1 are also related to Mg2 and Mg5 by the mirror plane at $z = \frac{1}{4}$. Four F atoms occupy Wyckoff symmetry positions 2(b), 2(d) or 2(e) above T_c , the remaining 20 F atoms occuping the 4(f) general positions.

It is noteworthy that four of the six *i*th atoms in SrMgF₄ have variable polar coordinates *above* T_c . In consequence, the uncertainty in Δz_i is propagated to that in the predicted magnitude of T_c . By contrast, each *i*th atom in a ferroelectric with symmetry higher than monoclinic generally undergoes displacement to *special* positions at or above T_c . In the case of ferroelectric YMnO₃ in space group $P6_3cm$, for example, $z'(Y1) \rightarrow \frac{1}{2}$, $z'(Y2) \rightarrow 0$ and $z'(Mn) \rightarrow \frac{1}{4}$ as $T \rightarrow T_c$ (Abrahams, 2001); no further uncertainty beyond that in the polar coordinates and the constant in (1) is thereby introduced in

the resulting Δz or T_c magnitudes in the case of higher symmetry ferroelectrics.

The most reliable measure of uncertainty in Δz_i for SrMgF₄ would be a comparison of the z' coordinates in Table 1 with an accurate experimental determination above T_c ; direct measurement of T_c would also allow an estimate of uncertainty. In the absence of such measurement, however, differences between Mg2, Mg3, Mg5 and Mg6 z coordinates may provide substitute indicators of uncertainty. Assuming these four Mg atoms have a common $z' = \pm 0.0828$ above T_c (see Table 1) leads to $\sigma \langle \Delta z_i(Mg) \rangle = 0.052$ Å; if the Mg atoms were to form independent pairs with $z' = \pm 0.0823$ and ± 0.0834 , then the mean uncertainty would be reduced to ~0.04 Å. The polar coordinates of the Mg atoms are, however, not necessarily confined to these alternatives. The uncertainty in Δz_i may hence differ strongly from either estimate.

Denoting octahedra by their Mg^{2+} ions, the centers of the F⁻ ion charge in Mg1(oct)–Mg6(oct) are located, respectively, at z' = 0.2569 (2), -0.0746 (2), 0.0908 (3), 0.2564 (3), -0.0746 (2) and 0.0923 (2). Each charge center is displaced further from the paraelectric position, corresponding to the z'(Mg) coordinates in Table 1, than the central Mg²⁺ ion and with the same sense. The effective displacement $\Delta z_i(Mg^{2+})$ is thus given by the octahedral displacement which ranges from 0.108 Å for Mg4(oct) to 0.150 Å for Mg6(oct). The Curie temperature in (1) necessarily corresponds to the largest Δz_i , hence T_c for SrMgF₄ is predicted to be 450 K with an uncertainty that may range between 370 and 270 K.

An anomaly at 1080 (10) K was observed by Bingyi & Banks (1982) for all compositions in the binary system $Sr_xMg_{1-x}F_4$ from x = 0.15 to 0.975. SrF_2 is cubic with a = 5.794 Å, MgF₂ is tetragonal with a = 4.628 and c = 3.045 Å, both structures differing considerably from that of SrMgF₄. Careful calorimetric and dielectric study of stoichiometric SrMgF₄ is necessary to determine if the variable-composition solid-state transformation noted is related to T_c .

3. Atomic displacement dimensionality and spontaneous polarization

The atomic displacements undergone by members of the ferroelectric BaCoF₄ family in the course of polarization reversal are strictly two-dimensional, all atoms in phase II (space group $A2_1am$) occupying a mirror symmetry plane (Keve *et al.*, 1970); the hypothetical phase I with $T_c > m.p.$ is expected to have the space group *Amam*. The corresponding atomic displacements to the paraelectric state in SrMgF₄ are substantial in all three dimensions, with the maximum value ~0.55 Å, see Table 1; they are, however, far from isotropic with average magnitudes 0.09, 0.23 and 0.23 Å along the *a*, *b* and *c* axes, respectively.

Ferroelectrics generally exhibit the highest P_s values, in the range ~ 10 to 70×10^{-2} C m⁻², for materials with atomic displacements under polarization reversal that are primarily one-dimensional. Intermediate magnitudes of P_s , ranging from ~ 3 to 10×10^{-2} C m⁻², correspond to atomic displacements that are primarily two-dimensional under ferroelectric

switching; ferroelectrics with displacements of comparable magnitude in each dimension have the lowest magnitudes with $P_s \lesssim 3 \times 10^{-2} \text{ Cm}^{-2}$ (Abrahams & Keve, 1971).

The magnitude and sense of P_s in SrMgF₄ may be derived from (2), using the atomic coordinates in Table 1 and a point charge model

$$P_s = (e/V)\Sigma Z_j \Delta z_j. \tag{2}$$

Summation over volume V for one unit cell containing j ions of charge Z_j (2+ for Sr and for Mg, 1– for F) displaced by Δz_j from the paraelectric arrangement, see Table 1, gives $P_s =$ 10.6 (6) $\times 10^{-2}$ C m⁻². This value is consistent with the distribution of atomic displacements in Table 1, both magnitude and dimensionality agreeing with the classification of SrMgF₄ as, primarily, a two-dimensional ferroelectric.

The strong polar properties of $SrMgF_4$ noted herein, together with the possibility of growing single crystals as large as the samples with ~18 mm diameter and 30 mm length reported by Ishizawa *et al.* (2001), offer the potential for use in a variety of pyroelectric, piezoelectric and nonlinear optical applications.

4. Phase transition nomenclature for SrMgF₄

The recommended nomenclature of Tolédano *et al.* (1998) for the hypothetical prototype phase I and predicted ferroelectric phase II of $SrMgF_4$, based on the results above, follows.

- I $| >450 | P112_1/m | Z = 12 | Nonferroic | Hypothetical$ (350) K | (11) | C = 12 | Nonferroic | Hypotheticalprototype phase
 - II $\begin{vmatrix} <450 \\ (350) \text{ K} \end{vmatrix}$ $\begin{vmatrix} P112_1 \\ (4) \end{vmatrix}$ $Z = 12 \begin{vmatrix} \text{Predicted} \\ \text{ferroelectric} \end{vmatrix}$ 2 variants

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In the paper by Abrahams (2002) Acta Cryst. (2002), B**58**, 34– 37 two errors were printed. First, in the abstract the Curie temperature was incorrectly given as 1450 K. The correct value is 450 K. The second error occurs in section (*a*) of p. 36. The polar displacement expression is currently given as $\Delta z_l \simeq 1$ Å. The correct expression should be $\Delta z_i \simeq 1$ Å.

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